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MATERIAL CHARACTERIZATION, PART A MECHANICAL PROPERTIES OF TWO METALS AT SEVERAL STRAIN RATES

S. A. EMERY

University of Dayton Research Institute Dayton, Ohio 45469

May 1982

Interim Report for period 1 October 1977 - 30 November 1979

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This report describes med	hanical proper	
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blade model is part of a fo		
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report contains a discussion conducted on two metallic m		
8Al-lMo-lV titanium. These		
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of their use in the J-79 blade and the F-101 blade respectively. Part B of the report contains a discussion of mechanical property tests conducted on the two composite components of a hybrid composite blade: boron/2024 aluminum and stainless steel wire mesh. These two composites are used in the hybrid composite APSI blade.

Static, quasi-static and dynamic tensile tests were conducted on the metallic materials because (1) studies indicate that metallic mechanical properties exhibit strain rate dependence and (2) tests conducted on full scale blade. show that various locations on the fan blades load at different rates. The mechanical properties measured include Young's Modulus (when obtainable), Poisson's ratio (when obtainable), the yield strength, the ultimate strength, and the ultimate strain. The density of the materials was also measured,

#### **FOREWORD**

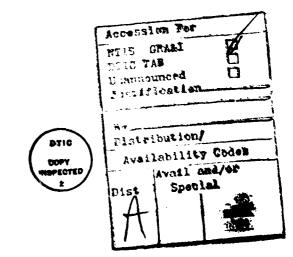
This report describes a contractual work effort conducted for the General Electric Company, Aircraft Engine Group under Purchase Order No. 200-FBA-14K-47844 which is a subcontract of F33615-77-C-5221.

This report covers work conducted during the period of October 1977 to November 1979 and is part of Task IV-A.

The GE Program Manager was Mr. Joe McKenzie and the Principal Investigator was Mr. Al Storace. The work reported hereir was performed under the direction of Susan A. Emery, Experimental and Applied Mechanics Division, University of Dayton Research Institute.

Technical support was provided by Mr. E. C. Klein. Program management for the University was provided by Mr. Robert Bertke.

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# SECTION I INTRODUCTION

The material characterization tests discussed in this report are part of a foreign object damage (FOD) study of jet engine fan blades. The tests constitute a portion of the gross structural damage subtask of that study. The FOD study is designed to provide two resources necessary for the successful development of FOD resistant fan blades: (1) an analysis capability for predicting jet engine response to FOD impacts and (2) an understanding of the mechanisms of FOD failures. The fan blade analytical model uses inputs such as mechanical property data (of the material under consideration) and impact loads (representative of ice balls, ice spears, or small birds) to predict the fan blade response to the foreign object damage impacts. The model allows considerable evaluation of the blade design and material prior to the manufacturing and testing of a full stage of blades. This eliminates many costly experiments that have been necessary in the past.

The material characterization tests provide mechanical properties for use in the blade model discussed above. The materials tested include two metals, 410 stainless steel and 8Al-1Mo-1V titanium (used in the J-79 blade and F-101 blade, respectively) and two composites, boron/2024 aluminum and stainless steel wire mesh (used in the hybrid composite APSI blade). Part A of this report contains the description of the material chracterization tests on the metals. Part B contains the discussion of the material chracterization tests on the composite materials.

The mechanical properties of metals usually exhibit some strain rate dependence.  $^{(1,2,3)}$  In addition, experiments have shown that the FOD impacts load various sections of the blades at different rates. For these reasons, tensile tests were conducted on the metals at strain rates ranging from  $10^{-3}$  to

about 10<sup>3</sup> strain/second. The quasi-static and intermediate rate tests (10<sup>-3</sup> to 1 strain/second), conducted with an MTS closed-loop hydraulic system, provide the following elastic and plastic region parameters: Young's modulus, Poisson's ratio (in the elastic region), 0.2 percent offset yield stress, ultimate stress, and ultimate strain. The dynamic tests (500 to 700 strain/second), conducted on a split-Hopkinson bar apparatus, provide only the plastic region parameters: yield stress, ultimate stress, and ultimate strain. The mechanical properties are summarized in this report as plots of the various parameters versus strain rate.

# SECTION II EXPERIMENTAL PROCEDURES

Tensile tests conducted at the strain rates shown in Table 1 provide the desired mechanical property data. This range of rates encompasses the range of strain rates measured at select locations on the full scale blade tests, 12-370 strain/second. (The full scale blade tests were conducted as part of the gross structural damage subtask IV-A but reported with similar tests in the Task VI report.) The quasi-static and intermediate rate tests (10<sup>-3</sup> to 1 strain/second) were run on an MTS electronly hydraulic closed-loop testing machine. They require different specimens and procedures than do the dynamic tests, which were run on the split-Hopkinson bar apparatus. Consequently, the two types of tests are described separately below.

TABLE 1
STRAIN RATES OF METALLIC MATERIAL
CHARACTERIZATION TESTS

Material	Strain Rates (strain/second)
8A1-1Mo-1V Titanium	10 <sup>-3</sup> , 10 <sup>-1</sup> , 1, 500
403 Stainless Steel	$10^{-3}$ , $10^{-1}$ , 1, 700

### 2.1 LOW AND INTERMEDIATE STRAIN RATE TESTS

The information obtained from the low and intermediate strain rate tests include two elastic parameters, Young's modulus and Poisson's ratio, and three plastic parameters, the yield stress, the ultimate stress, and the ultimate strain. The following paragraphs describe the specimens used and how they were instrumented and tested. The full true stress-true strain curves appear in Appendix A.

## 2.1.1 Test Specimen and Specimen Instrumentation

The specimens were made from bar stock which had been worked the same amount as the finished blades. The specimen is illustrated in Figure 1. Its dimensions were per ASTM standard E8-69. Three specimens of each material were tested at each rate with one exception. Only one specimen of the titanium was tested at 1 strain/second.

mentation placed on the specimens. The extensometer supplied information for obtaining the entire stress-strain from test start to specimen failure. The high resistance foil strain gage placed in the longitudinal direction (the same direction as the applied load) provided high resolution measurements for the parameters in the elastic region and at yield. The strain gage in the transverse direction (to the applied load) supplied measurements for the calculation of Poisson's ratio. The transverse strain gages were applied to only one specimen of each material at each strain rate. The data reduction equations appear in the analysis section of this report.

### 2.1.2 Test Procedures and Test System

The following paragraphs contain a brief description of the test system and test procedures used for the low and intermediate strain rate tests.

## 2.1.2.1 Test System

The low and intermediate strain rate tests were conducted using the University's MTS electrohydraulic closed-loop testing system. This closed-loop test machine can be programmed in load, strain, or displacement control. The machine can produce displacement rates that range from less than 0.05 in/min. to 400 in/min. The signal conditioner used with the strain gage was a Honeywell 218 Bridge Amplifier. Included in the test system is an X-Y recorder and a multichannel transient recorder. An additional two pen X-Y recorder was used for these tests.

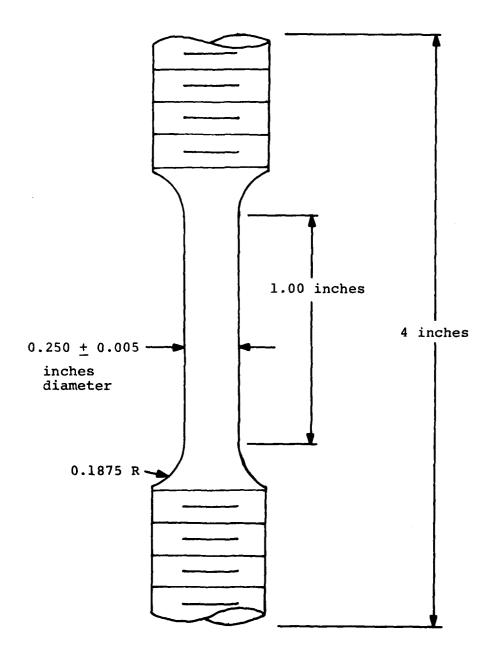
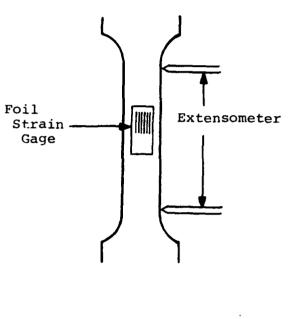
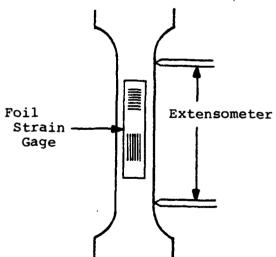


Figure 1. Specimen for Low and Intermediate Strain Rate Tests.



Specimen instrumented for axial strain and displacement measurement



Specimen instrumented for axial strain and displacement measurements and transverse strain measurements

Figure 2. Location of Measurement Instrumentation on the Low and Intermediate Strain Rate Specimens.

#### 2.1.2.2 Test Procedure

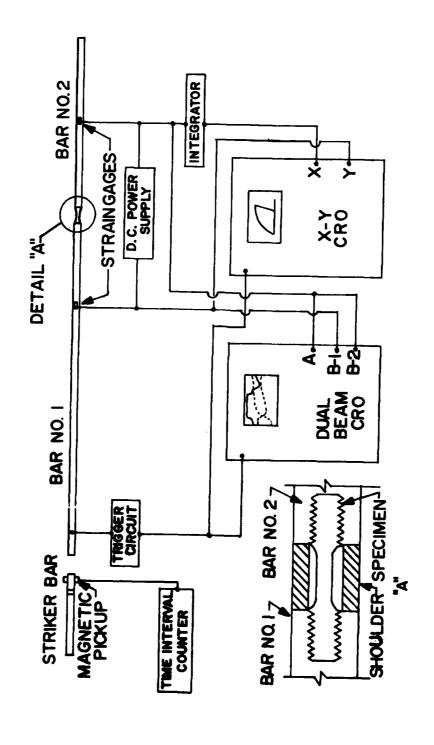
After tightening the strain gaged specimen in the grips, the extensometer was attached and adjusted to give the proper output voltage for the desired gage length. With the test machine in the load control mode a linear ramp function loaded the specimen to failure. Load and displacement were recorded directly on the system X-Y recorder for the lowest strain rate tests (10<sup>-3</sup> strain/second). Load and strain were recorded directly on the additional X-Y recorder. For the higher strain rate tests (10<sup>-1</sup> and 1 strain/second) the three parameters, load, strain, and displacement were recorded on three channels of the transient recorder for playback at slower speeds into the X-Y recorder.

#### 2.2 HIGH STRAIN RATE TESTS

Mechanical properties obtained from the high strain rate tests include the following three plastic parameters: the yield stress, the ultimate stress, and the ultimate strain. Plastic region true stress-true strain curves were also acquired from the tests. (Elastic properties cannot be obtained from these tests because the pressure waves in the Hopkinson bar are not considered to be in equilibrium in the elastic region). The method (and apparatus) used is the same as that used in a study of the high strain rate tensile mechanical properties study of beryllium. (5) The subsequent paragraphs provide a detailed description of the experimental apparatus, the instrumentation, and the test procedures and mechanisms. The analysis section of this report contains the brief explanation of the data reduction process. The results appear in Appendix B.

### 2.2.1 Experimental Apparatus

A schematic of the split Hopkinson bar apparatus and the associated instrumentation appears in Figure 3. Three parts of the apparatus are of major interest: two Hopkinson pressure bars (bar No. 1 and bar No. 2) and the striker bar.



Schematic of Apparatus and Instrumentation for High Strain Rate Tests. Figure 3.

Bar No. 1 must be twice the length of bar No. 2 and the striker bar must be shorter than bar No. 2. For these experiments the bars are 0.06, 0.15, and 0.30 m lengths of 12.7 mm diameter AISI 4130 steel. They are mounted and aligned on a rigid base.

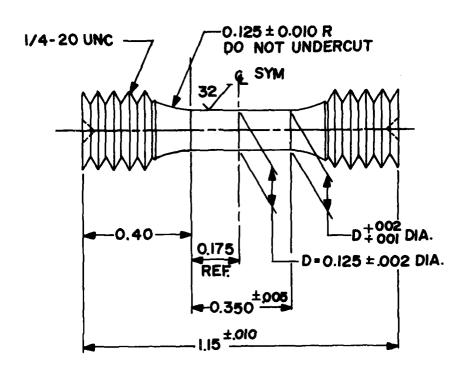
Detail "A" of Figure 3 shows an enlargement of a longitudinal section of the apparatus at the location of the specimen. To prepare for the test, the threaded tensile speciment, shown in Figure 4, is screwed partially into the pressure bars. Then a split shoulder is placed over the specimen, and the specimen is screwed in until the shoulder is snug against the pressure bars. The shoulder has the same outer diameter as the pressure bars (12.7 mm) and it has an inner diameter of 6.4 mm, just large enough to clear the specimen. The shoulder is made of the same material as the pressure bars, AISA 4130 steel. The ratio of the cross-sectional area of the shoulder to that of the pressure bar is 3:4, while the ratio of the area of the shoulder to the net cross-sectional area of the specimen is 12:1.

### 2.2.2 Instrumentation

The recording circuitry consists primarily of strain gages, two oscilloscopes, and a counter. See Figure 3. The recording system is triggered when the striker bar impacts bar No. 1. A brief review of the major parts of the recording system follows.

Bars No. 1 and 2 are instrumented with high resistance foil strain gages. The gages on each bar are placed equidistant from the specimen so that the reflected and transmitted wave signals are time coincident. They are far enough from the specimen that no spurious reflections interfere with the pulses being recorded. Gages are placed diametrically opposite each other on the pressure bars to cancel bending.

The two oscilloscopes record data. One, a dual beam Tektronix type 565 oscilloscope, records the complete



NOTE: ALL DIMENSIONS IN INCHES

## THREADED TENSION HOPKINSON BAR SPECIMEN

Figure 4. Specimen for High Strain Rate Tests.

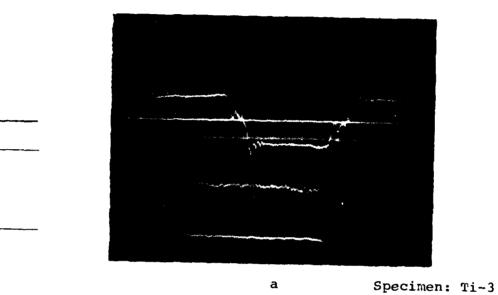
strain time history of the test. One beam records the incident pulse as it passes the first gage. The second beam is used in chopped mode to record both the transmitted pulse (which is proportional to load) and the reflected pulse (which is proportional to strain rate). Figure 5(a) shows a photo of typical traces.

The second oscilloscope, an X-Y Tektronix, records a load-deflection curve. The reflected pulse (from gage No. 2) passes through an electronic integrator and is fed into the X axis of the scope. The signal is proportional to the displacement of the specimen. The signal from gage No. 1, proportional to the load in the specimen, is fed to the Y axis. Figure 5(b) shows the load-deflection trace.

## 2.2.3 Test Procedures and Mechanisms

The experiment begins when the striker bar is accelerated so that it impacts bar No. 1. The compression pulse generated travels down bar No. 1 to the specimen-shoulder junction. The pulse passes almost exclusively through the shoulder because of (1) the large area ratio of shoulder to specimen mentioned above and (2) the loose fit of the threaded joints (specimen and pressure bars). The portion of the pulse that passes through the specimen is below the elastic limit of the material. Then the compressive pulse travels through bar No. 2 to the free end, where it reflects as a tensile pulse. When this reflected pulse reaches the specimen, part of it is transmitted through the specimen and part is reflected back into bar No. 2 (because the shoulder doesn't carry the tensile pulse). The incident, transmitted and reflected pulses are recorded and analyzed for the required information.

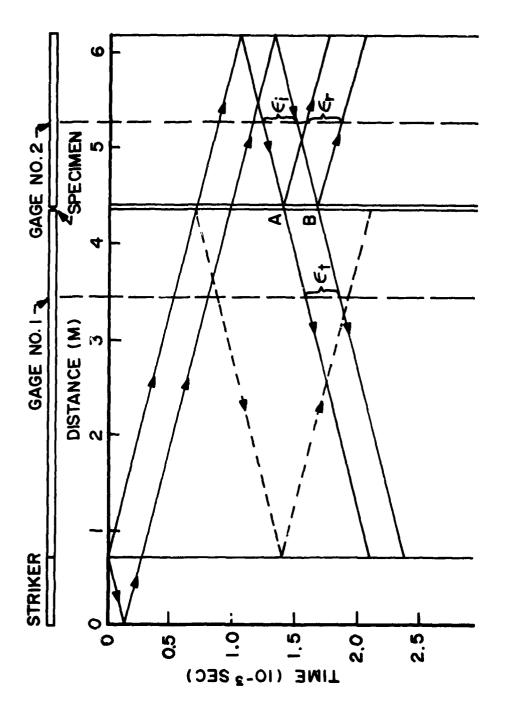
Figure 6 shows a Lagrangian x-t plot which illustrates the path of the pressure pulse during the test. The amplitude of the incident pressure pulse, generated by the striker bar impacting bar No. 1 is dependent on the striker bar velocity. Its length is twice the longitudinal elastic wave



b

Vert: 465 lb/div.
Horiz:
 0.00155 in/div.

Figure 5. Typical Oscilloscope Traces.



Lagrangian Displacement - Time Plot for High Strain Rate Tests. Figure 6.

transit time in the striker bar. The plot shows that the compressive pulse passes through the entire apparatus and is reflected from the free end of bar No. 2 as a tensile pulse,  $\epsilon_{\bf i}$ . The plot also illustrates the tensile pulse splitting when it reaches the specimen-shoulder junction and part of it,  $\epsilon_{\bf t}$ , being transmitted through the specimen while part,  $\epsilon_{\bf r}$ , is reflected back into bar No. 2.

# SECTION III ANALYSIS

This section outlines the data reduction techniques and equations used to obtain the desired true stress-true strain curves and specific elastic and plastic parameters. All of the test data began as analog load-displacement curves that were then analyzed by various computer routines particular to the test type (low and intermediate, or high strain rate). The results appear in the appendices in analog and digital form.

#### 3.1 LOW AND INTERMEDIATE STRAIN RATE TESTS

The low and intermediate strain rate tests resulted in (1) load-displacement curves from the extensometer and (2) load-strain curves from the strain gages. Converting the displacement measurement to true strain is relatively simple. Using u as displacement and  $\ell_{\rm O}$  as the initial gage length, one obtains the following:

$$\varepsilon_{\rm E} = \frac{\rm u}{\ell_{\rm O}}$$
 (engineering strain) (1)

$$\varepsilon_{\rm T} = \ln (1 + \varepsilon_{\rm E})$$
 (true strain) (2)

where ln is the natural logarithm.

The extensometer gage length,  $\ell_{\rm O}$ , was 1 inch for these tests. Consequently, the digitized displacement value could be used directly in calculations as engineering strain.

The strain gage data was also simple to reduce. A correction factor was applied to the strain reading to account for the nonlinearity of the bridge circuit. The incremental error, n, is added to the indicated,  $\hat{\epsilon}$ , to obtain the actual strain,  $\epsilon$ , causing the resistance change in the gage. For this case, a single active gage in a quarter-bridge arrangement, the

correction is

$$n = \frac{F(\hat{\epsilon})^2 \times 10^{-6}}{2 - F(\hat{\epsilon}) \times 10^{-6}}$$
 (3)

where F is the gage factor. Equation 2 is then used to obtain true strain.

Obtaining true stress values for the low and intermediate strain rate tests is more complicated. True stress,  $\sigma_{\mathbf{T}}$ , is proportional to engineering stress,  $\sigma_{\mathbf{E}}$ , by the ratio of the initial area,  $\mathbf{A}_{\mathbf{O}}$ , to the current area,  $\mathbf{A}_{\mathbf{O}}$ . In the elastic region, where one has small strains, the ratio of areas can be expressed in terms of the longitudinal engineering strain so that

$$\sigma_{\rm T} = \frac{\sigma_{\rm E} (1 + \varepsilon_{\rm E})}{1 + \varepsilon_{\rm E} (1 - 2\nu)}$$
 (elastic region true stress) (4)

In the plastic region one assumes the material is incompressible, consequently the volume remains constant. Based on this assumption the true stress in the plastic region becomes

$$\sigma_{\mathbf{T}} = \frac{\sigma_{\mathbf{E}} (1 + \varepsilon_{\mathbf{E}})}{1 + \bar{\varepsilon}_{\mathbf{p}} (1 - 2\nu)}$$
 (plastic region true stress) (5)

where  $\bar{\epsilon}_{E}$  is the engineering strain evaluated at the yield point of the material.

Three elastic region parameters were calculated from the strain gage data. Young's modulus is the slope of the linear part of the curve. Poisson's ratio,  $\vee$ , was obtained for one test per material per strain rate with the following relationship:

$$v = +\varepsilon_{\text{transverse}}/\varepsilon_{\text{longitudinal}}$$

The shear modulus, G, was calculated with the equation

$$G = \frac{E}{2(1 + v)}$$

The values of these mechanical properties appear in the results section. An additional parameter, the yield stress, was obtained from these curves. It was evaluated using the 0.2 percent (strain) offset method because neither of the metals exhibits a definite yield point. The load-displacement curves were analyzed using Equations 2,4, and 5 to produce the full true stress-time strain curves. The ultimate strain was taken as the final strain value of the curve. However, the ultimate stress was calculated using the final load divided by the final reduced area of the specimens because Equation 5 is not accurate (complete) for large amounts of plasticity.

These curves were all digitized on the University of Dayton's Tektronix 4014-1 computer graphics terminal and associated digitizing board.

#### 3.2 HIGH STRAIN RATE TESTS

The polaroid photographs of the load-deflection traces obtained from the high strain rate tests were digitized using a Hewlett Packard 9800 series calculator and digitizer (located at the AFML) to produce true stress-engineering strain curves. Those curves were then digitized on the Tektronix 4014-1 computer graphics terminal and digitizing board (located at the University) to produce true stress-true strain curves. A brief review of the analysis in the two data reduction programs follows. Details of the theory of the measurements appear in documents by U. S. Lindholm (6) and by Lindholm and L. M. Yeakley. (7)

The average stress in the specimen is:

$$\sigma_{s} = E (A/A_{s}) \epsilon_{t}$$
 (6)

where E and A are the elastic modulus and cross-sectional area of the pressure bars,  $A_s$  is the cross-sectional area of the specimen and  $\epsilon_t$  is the transmitted wave. Using the assumption of no volume change after first yield, Equation (6) is modified to:

$$\sigma_{\mathbf{T}} = E(\mathbf{A}/\mathbf{A_s}) \quad \varepsilon_{\mathbf{t}} \quad \frac{1}{(1+\varepsilon_{\mathbf{t}})}$$
 (7)

for true stress.

Engineering strain was obtained from the deflection values using the following relationship:

$$\varepsilon = 0.265 \delta - 0.5 (1 - \exp^{-0.55 \delta})$$

where  $\epsilon$  represents strain in percent and  $\delta$  represents the deflection in mils between the grips. (5) This equation results from least-squares fitting many deflection-strain plots from several different materials. It is particular to the specimen size used, not to a material.

The yield stress was determined as the point on the stressstrain curve at which the increase in applied load essentially ceased. The ultimate stress and ultimate strain were selected as the final point on the stress-strain curve. Reasons for this selection procedure appear in the results and discussion section.

# SECTION IV RESULTS AND DISCUSSION

The mechanical properties acquired from the various strain rate tests on 8Al-lMo-lV titanium and 410 stainless steel appear in Tables 2 and 3, respectively. Table 4 shows the measured density of each material. Figures 7 through 10 are plots of the mechanical properties versus strain rate. The points lacking standard deviation marks either represent one data point or they represent averaged data having a standard deviation smaller than the data symbol. The true stress-true strain curves for the low and intermediate tests (Appendix A) have two parts to the curve in the plastic region. The solid line represents the test data analyzed with the plastic region equations given in the analysis section. Obviously these are not complete for large strains. The dotted line joins the ultimate stress (calculated with the final load and reduced cross-sectional area)-ultimate strain point to the test curve.

Studying Figures 7 through 10, one concludes that all of the tensile mechanical properties determined by these tests vary with strain rate with the exception of the elastic modulus of titanium. Perhaps additional testing of the stainless steel would reveal that its elastic modulus also does not vary with strain rate. It is important to consider the comments in the following paragraphs when using the curves shown as data for calculations.

A very small specimen must be used for high strain rate tests to satisfy assumptions made in the Hopkinson bar theory equations. This presents problems in obtaining accurate data from the tests, particularly in the elastic region. One problem is that achievable machining tolerances on the specimen in relation to load surface alignment make it difficult to resolve small strains accurately. Secondly, an assumption made in developing the Hopkinson bar equations is that many stress

TABLE 2
MECHANICAL PROPERTIES OBTAINED FOR 8A1-1MO-1V TITANIUM

Specimen	Strain Rate (E/sec)	Elastic Modulus (x 10 <sup>6</sup> psi)	Shear Modulus (* 10 <sup>6</sup> nei)	Yield Stress	Ultimate Stress	Ultimate Strain
		(==== == :=)	(+04 o+)	(Tev)	(TSY)	(bercent)
l-Ti	0.001	16.9	6.5	148.9	202.5	28.2
2-Ti	0.001	17.3	6.7	148.7	200.8	27.2
3-Ti	0.001	17.4	8.9	148.7	199.3	21.8
4-Ti	0.10	15.8	6.3	156.3	188.5	13.7
5-Ti	0.10	16.9	8.9	153.0	189.1	15.9
6-Ti	0.10	17.1	8.9	156.4	194.5	15.3
7-Ti	1.0	17.3	8.9	164.0	184.6	15.7
Ti-3	550	*	*	203.0	161.1	13.7
Ti-2	260	*	*	202.0	154.2	14.5
Ti-1	580	*	*	208.0	158.7	15.3
		-				

Poisson's Ratio = 0.279

\* Elastic properties are not obtainable from the high strain rate tests.

TABLE 3
MECHANICAL PROPERTIES OBTAINED FOR 410 STAINLESS STEEL

Specimen	Strain Rate (E/sec)	Elastic Modulus (x 10 <sup>6</sup> psi)	Shear Modulus (x 106 psi)	Yield Stress (ksi)	Ultimate Stress (ksi)	Ultimate Strain (percent)
4-SS	0.001	27.4	10.6	84.2	207.2	26.7
5-55	0.001	30.5	11.7	84.8	207.2	27.2
6-55	0.001	33.7	12.9	84.9	211.9	26.9
85-6	0.10	27.7	11.2	87.2	191.9	25.7
10-55	0.10	26.4	10.2	87.0	188.3	26.8
11-55	0.10	28.2	10.9	89.2	198.3	23.5
12-SS	1.0	27.2	10.4	9.98	182.7	23.8
13-SS	1.0	26.1	10.0	91.3	185.5	25.4
14-SS	1.0	27.2	10.4	94.5	183.8	24.6
15-55	1.0	28.0	10.9	88.8	184.7	25.6
SS-5	0.089	*	*	124.0	129.5	16.9
SS-4	760.0	*	*	123.0	130.0	17.2
SS-3	700.0	*	*	131.0	131.4	18.0
SS-2	530.0	*	*	128.0	*	*
SS-1	340.0	*	*	126.0	*	* *

Poisson's Ratio = 0.297

\* Elastic properties are not obtainable from the high strain rate tests.

\*\* These specimens did not fail.

TABLE 4
DENSITY

Material	Density
8A1-1Mo-1V Titanium	4.32 g/cm <sup>3</sup>
410 Stainless Steel	7.57 gm/cm <sup>3</sup>

wave reflections occur in the specimen before a state of dynamic equilibrium exists. This does not happen until some plastic deformation has taken place. In addition, once necking occurs in the specimen a uniform stress field no longer exists making the equations for stress and strain invalid. This is a significant factor for the small specimens.

However, the split-Hopkinson bar apparatus represents the state-of-the-art technique for obtaining stress-strain information at high strain rates. The curves are considered most accurate in the range of 2 to 10 percent strain. The yield stress values reported represent the point on the curves where the load ceases to increase significantly for a rather large increase in displacement. (The 0.2 percent offset method obviously cannot be used because of the lack of data in the elastic region.) The ultimate stress and ultimate strain values from these tests were the final point of the curve. Although the values from the high strain rate tests cannot be used as exact data points they are useful in establishing the trend of the strain rate dependence of the subject materials over nearly seven decades of strain rate.

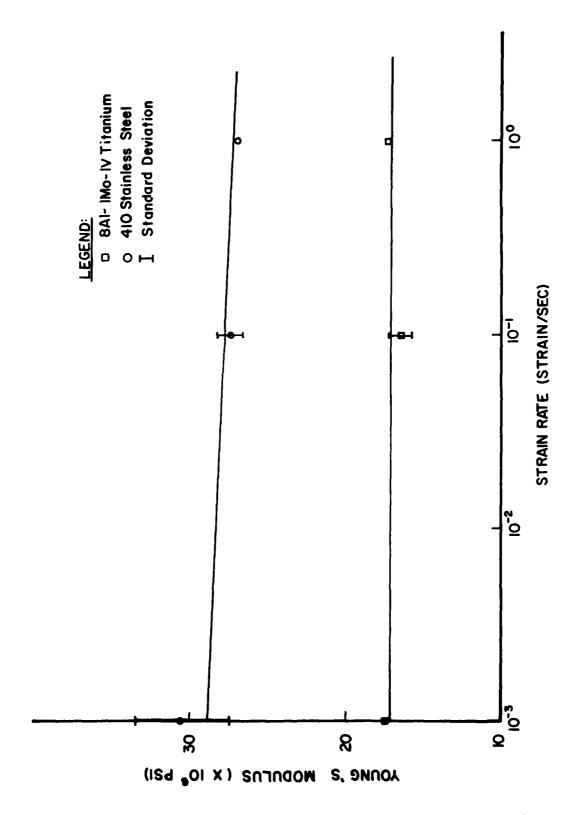


Figure 7. Young's Modulus VS. Strain Rate.

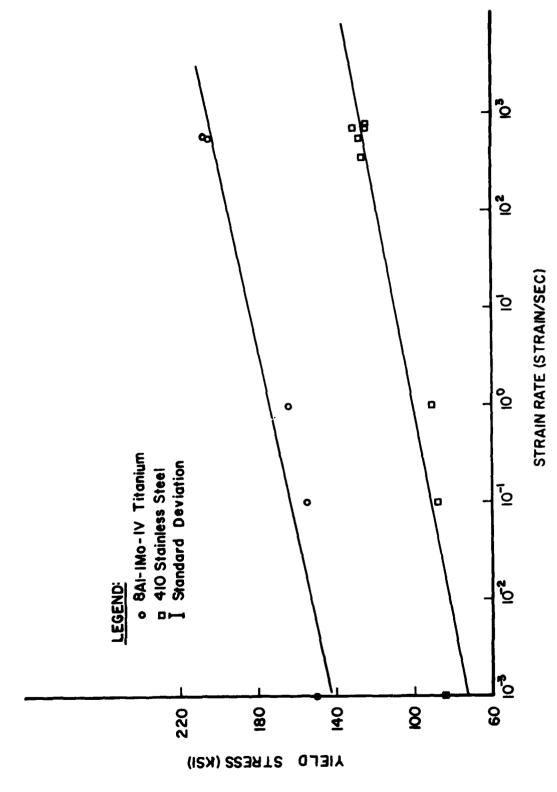


Figure 8. Yield Stress VS. Strain Rate

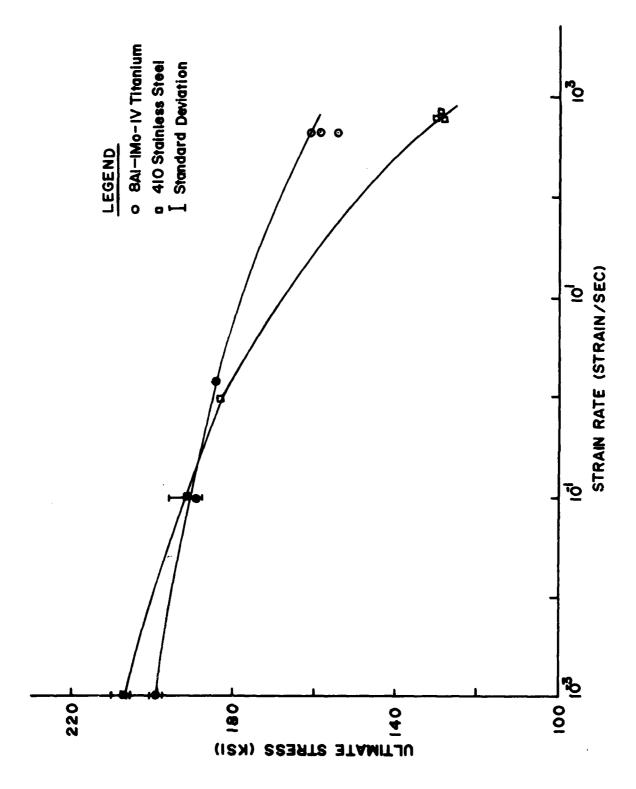


Figure 9. Ultimate Stress VS. Strain Rate.

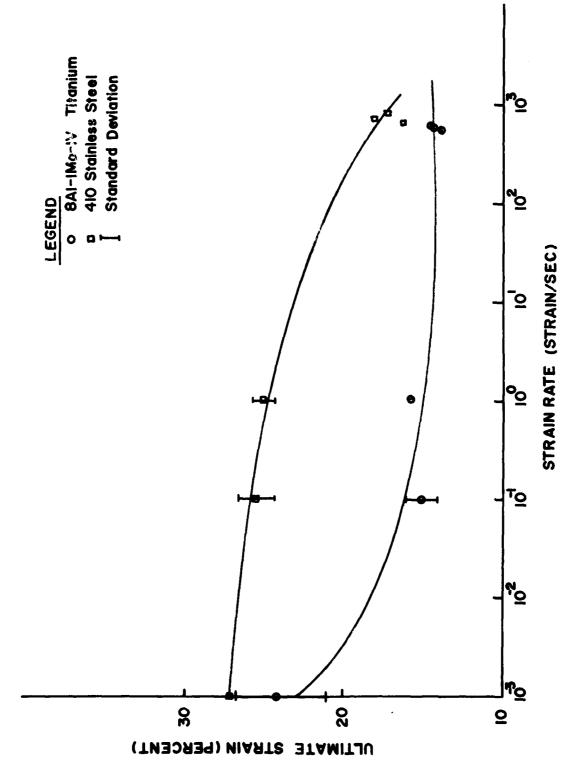


Figure 10. Ultimate Strain VS. Strain Rate.

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- 2. Green, S.J. and S.G. Babcock, "Response of Materials to Suddenly Applied Stress Loads: Part I: High Strain-rate Properties of Eleven Reentry-vehicle Materials at Elevated Temperatures," TR66-83 Part I, November 1966.
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- 4. 1977 Annual Book of American Society for Testing and Materials Standards, Part 10: Metals Physical, Mechanical, Corrosion Testing, pp. 154-173.
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- 6. Lindholm, U.S., "Some Experiments with the Split Hopkinson Pressure Bar," Journal of the Mechanics and Physics of Solids, 1964, Vol. 12, pp. 317-335.
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## APPENDIX A

LOW AND INTERMEDIATE STRAIN RATE CURVES AND DIGITAL DATA

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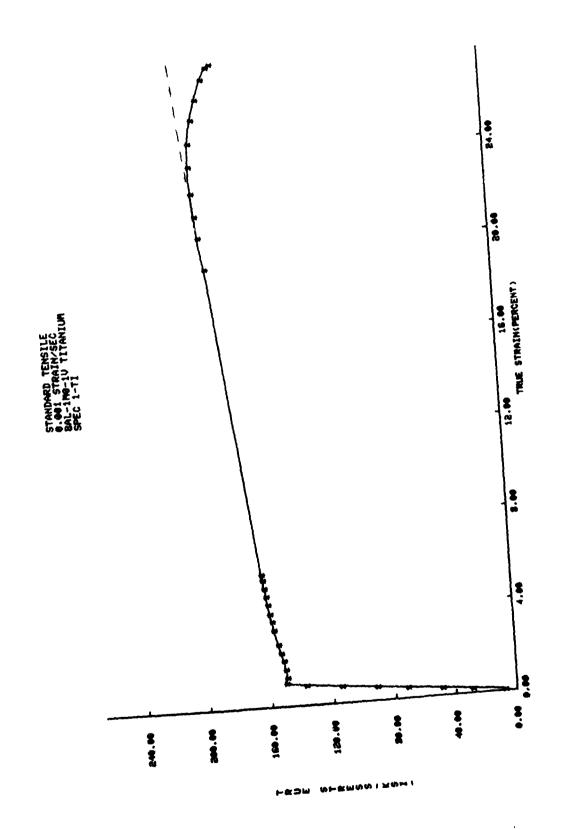
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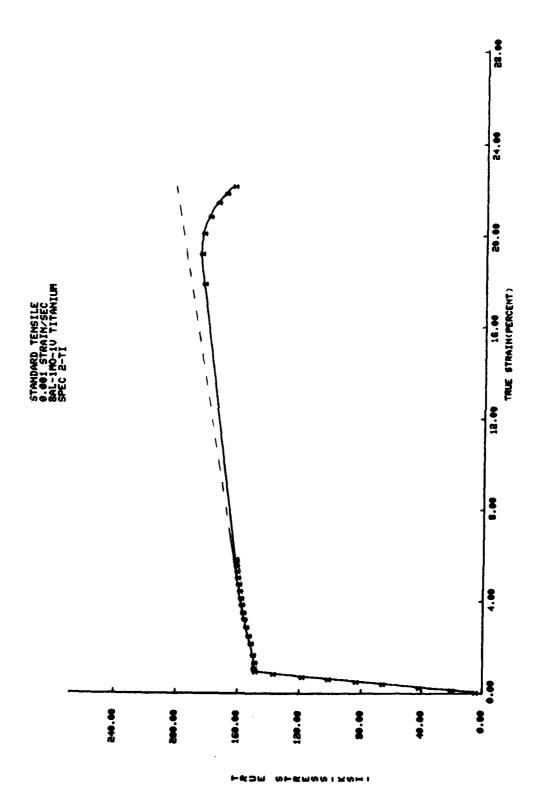
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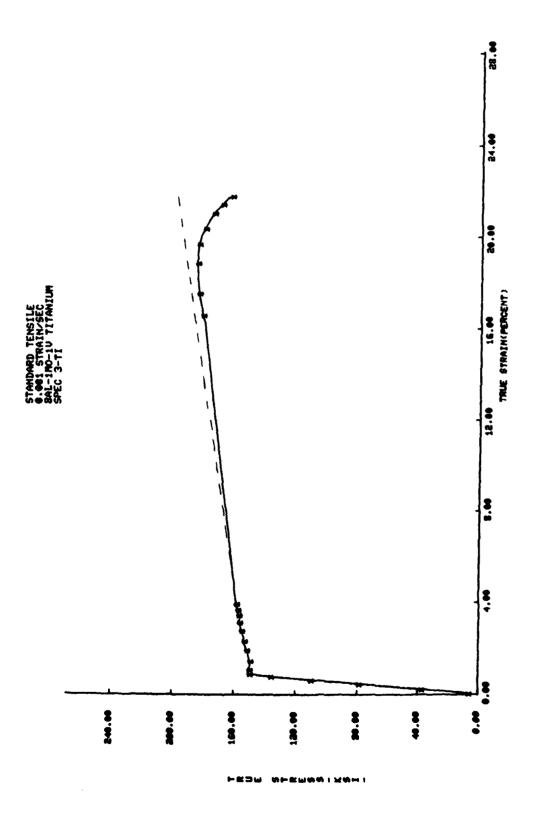
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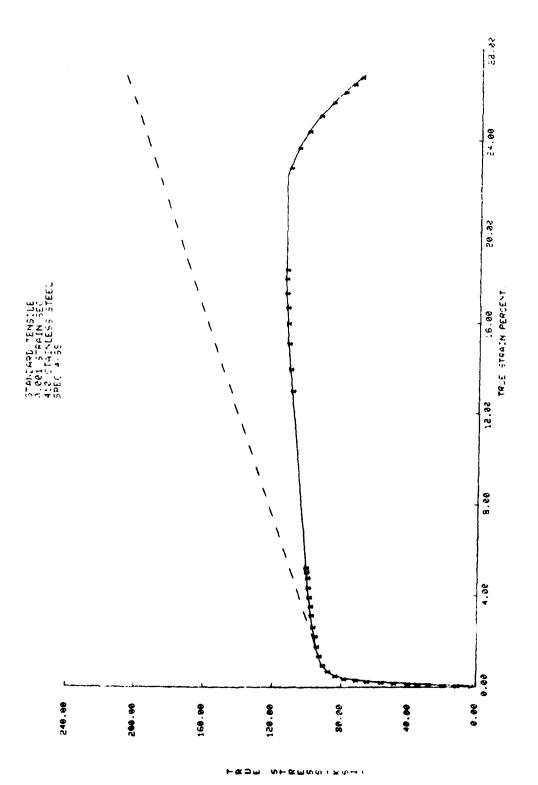
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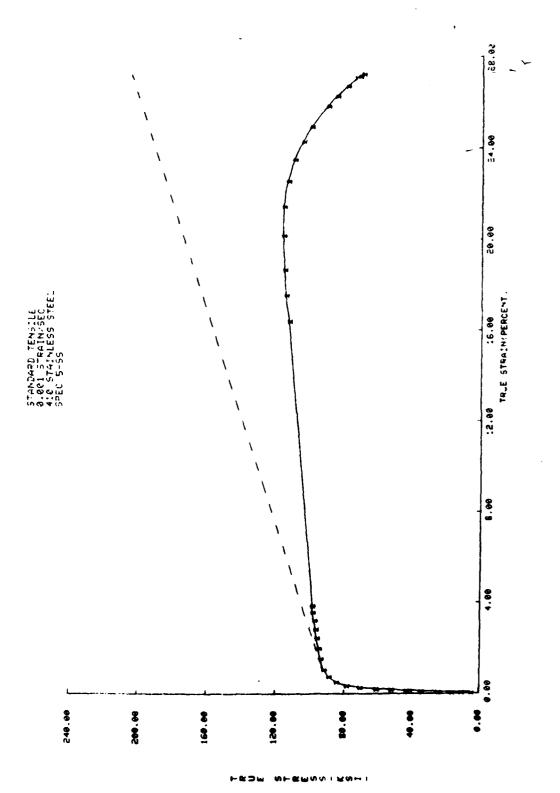
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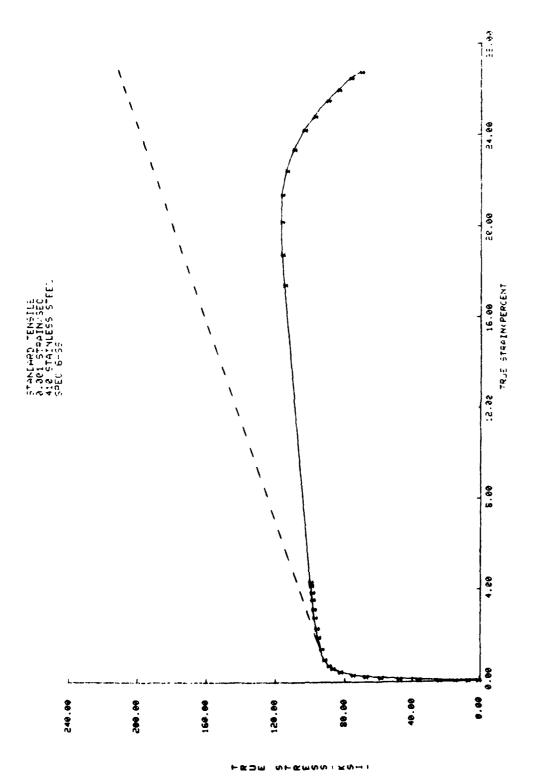












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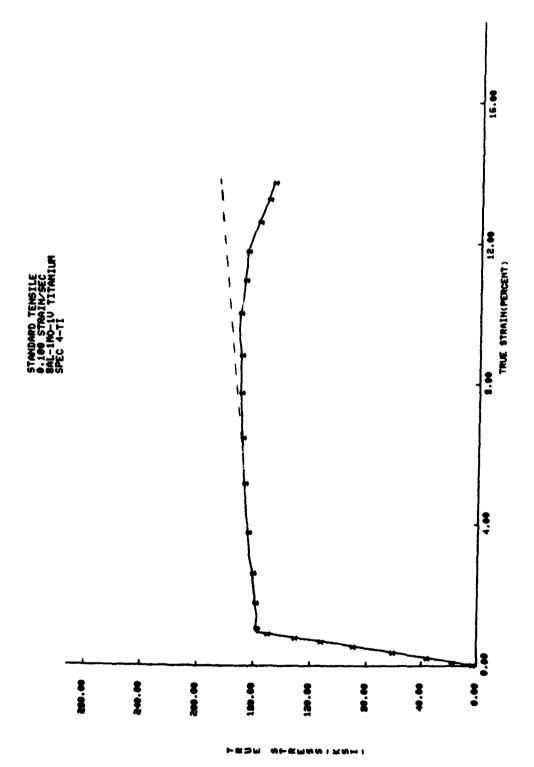
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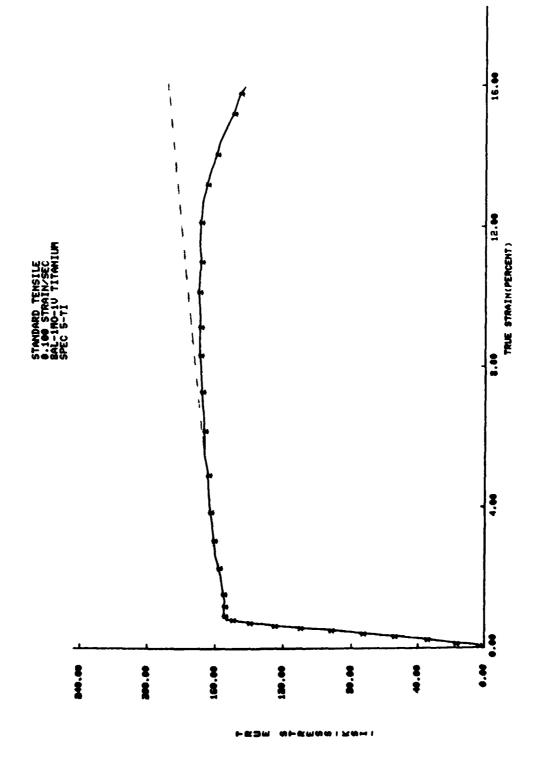
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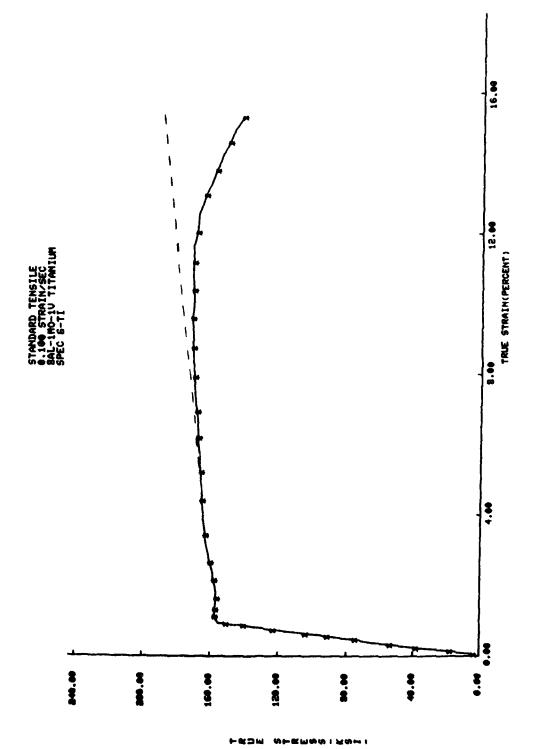
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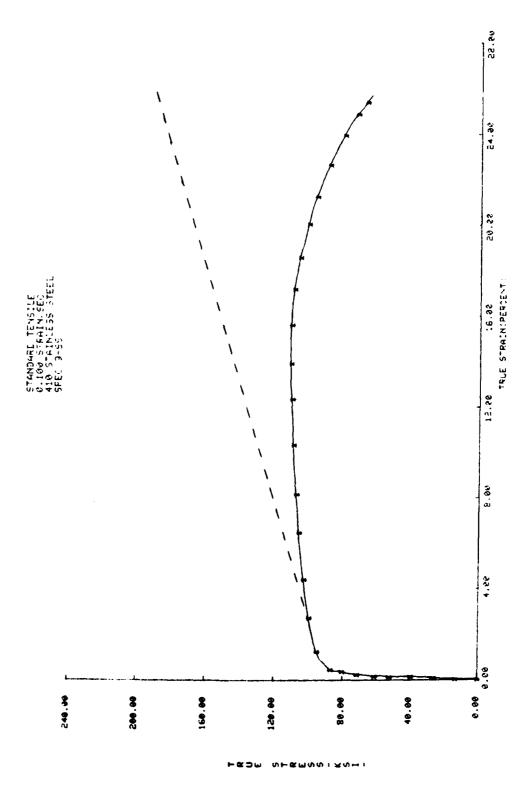
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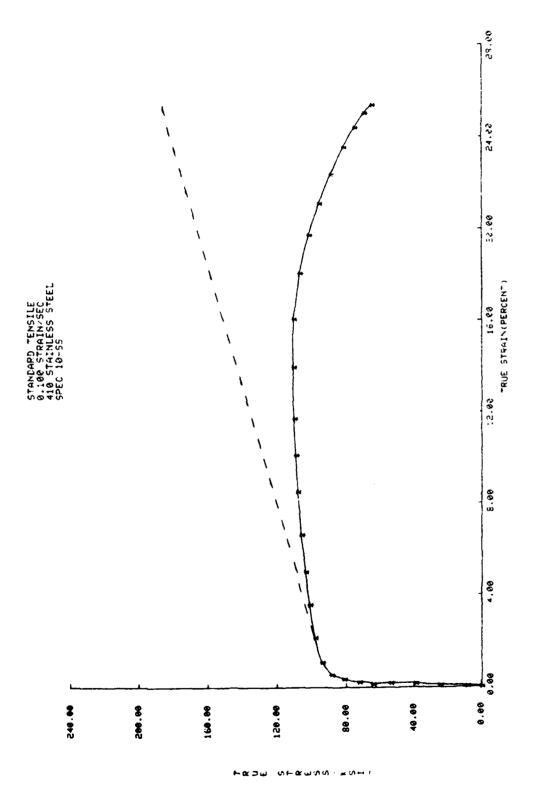
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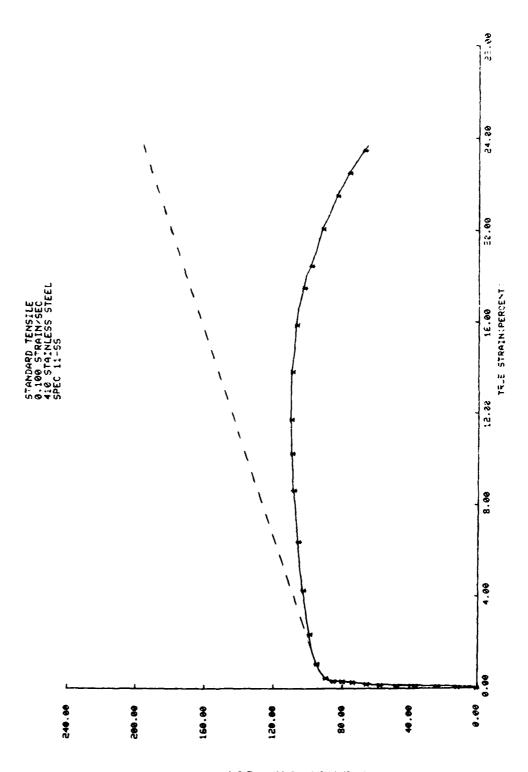








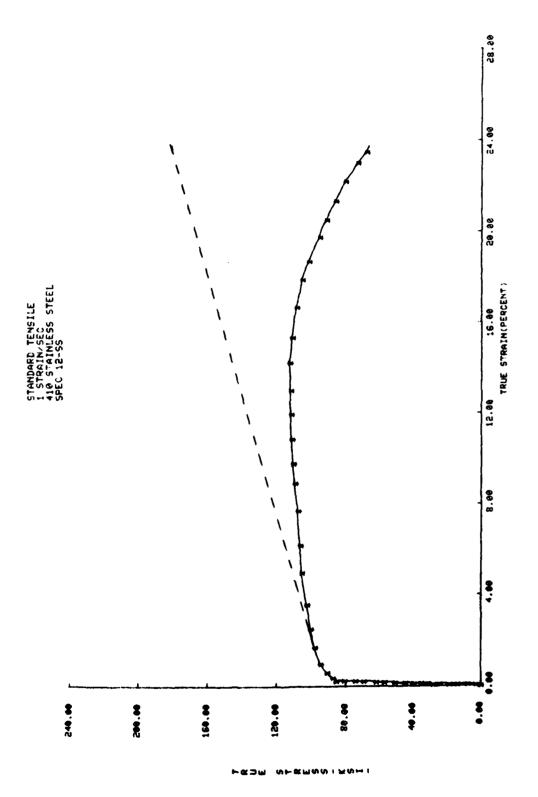


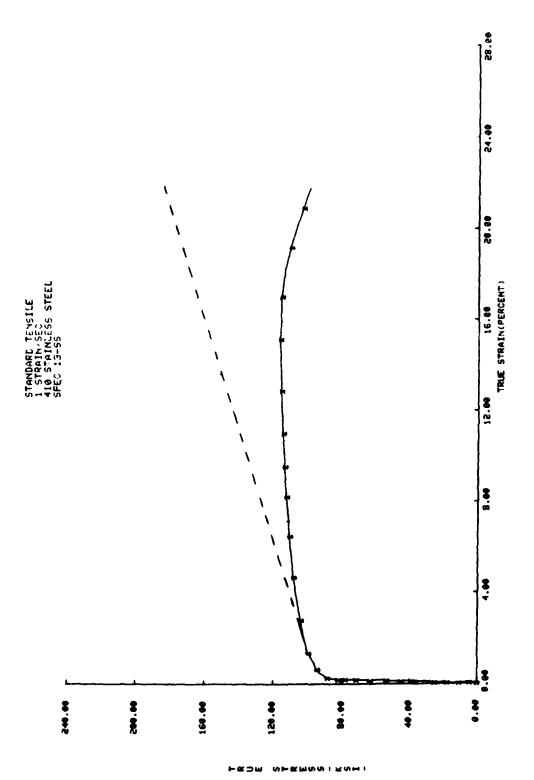


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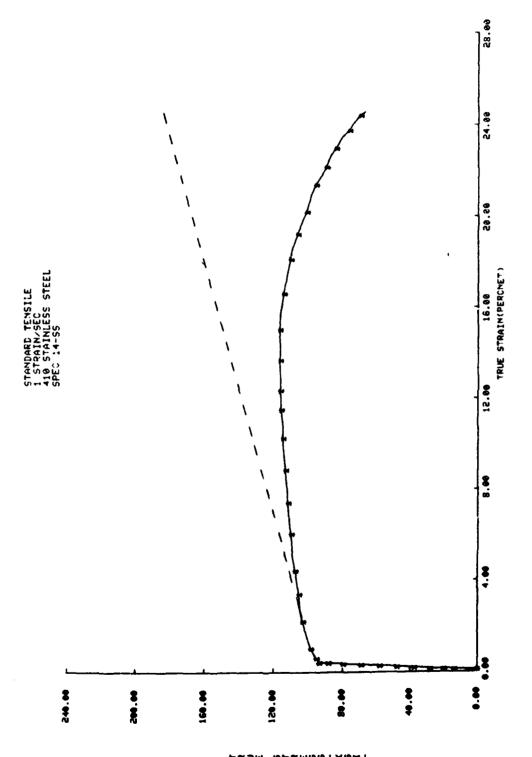
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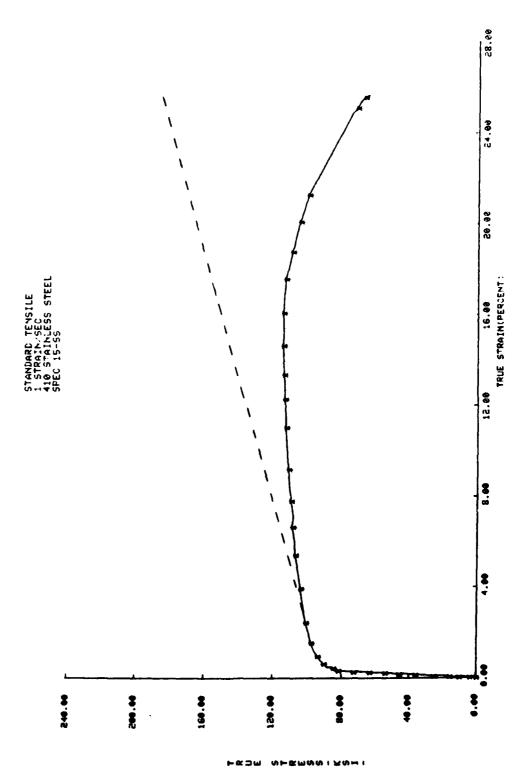


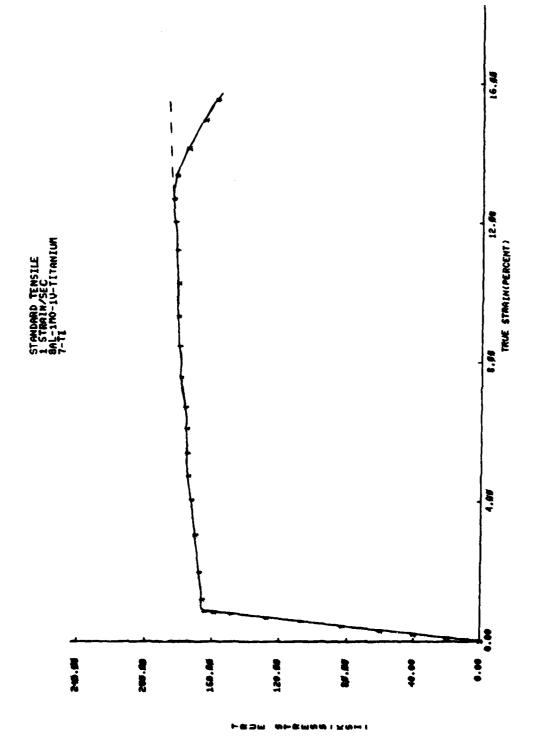
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## APPENDIX B

HIGH STRAIN RATE CURVES AND DIGITAL DATA

